

Computer modeling of crystalline electrolytes: Lithium thiophosphates and phosphates

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Solid Electrolytes

(compared to liquid electrolytes)

Advantages

- Physically and chemically stable
- Low electronic conductivity (prevents self-discharge)
- Performs well in thin film batteries

Disadvantages

- Poor contact area for high capacity electrodes
- Interface stress for electrode with volume changes
- Lower conductivity per area.

Thiophosphates vs. Phosphates

LiPON
(Oak Ridge National Lab)

$$\sigma \sim 10^{-6} \text{ S/cm}$$

$$E_A \sim 0.3-0.7 \text{ eV}$$

$\text{Li}_2\text{S}-\text{P}_2\text{S}_5$
(Osaka University)

$$\sigma \sim 10^{-3} \text{ S/cm}$$

$$E_A \sim 0.1-0.3 \text{ eV}$$

Ref:

-N. J. Dudney, Interface 17 (3), 44 (2008).

-J. B. Bates, N. J. Dudney, B. Neudecker, A. Ueda, and C. D. Evans, Solid State Ion. 135, 33 (2000).

Ref:

-A. Hayashi, K. Minami, F. Mizuno and M. Tatsumisago, J. Mater. Sci. 43 (2008), pp. 1885–1889.

-M. Tatsumisago and A. Hayashi, J. Non-Cryst. Solids 354 1411-1417 (2008)



Comparison of various $Li_2S-P_2S_5$ derivatives.



From: M. Tatsumisago and A. Hayashi, J. Non-Cryst. Solids 354 1411-1417 (2008)

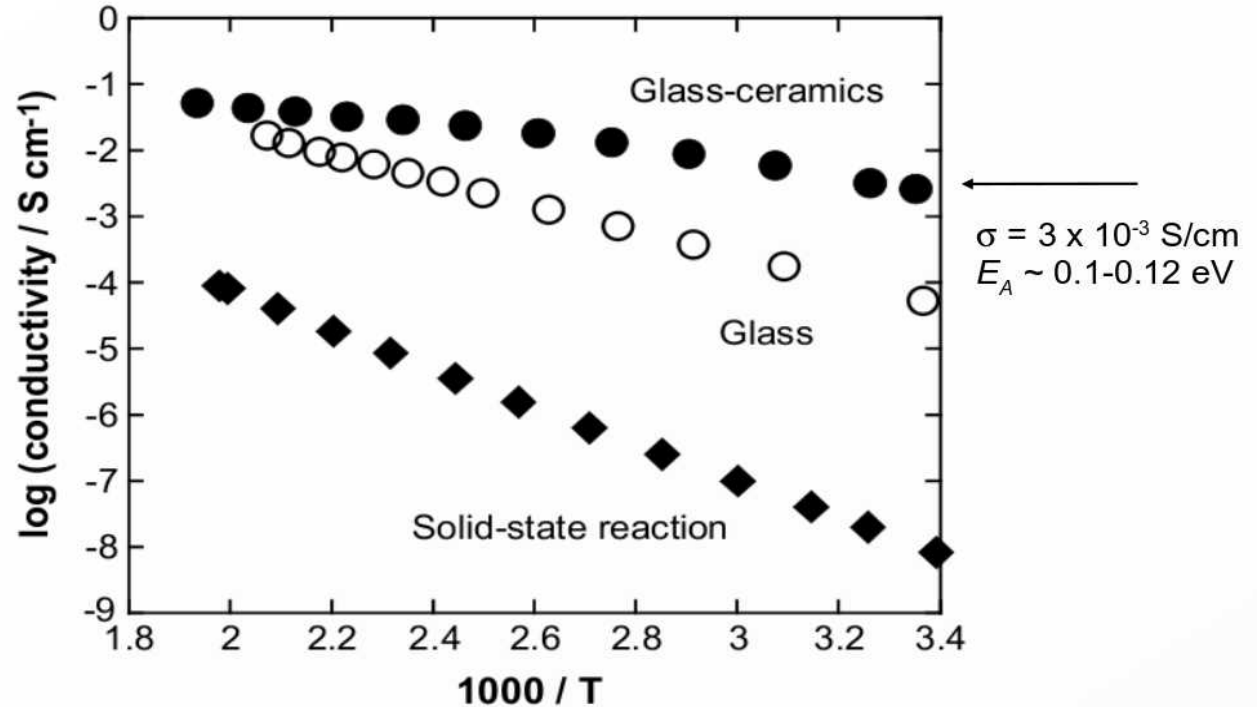


Fig. 5. Temperature dependences of the conductivities for the $70Li_2S \cdot 30P_2S_5$ glass and glass-ceramics. The conductivity data for the sample prepared by solid-state reaction are also shown.

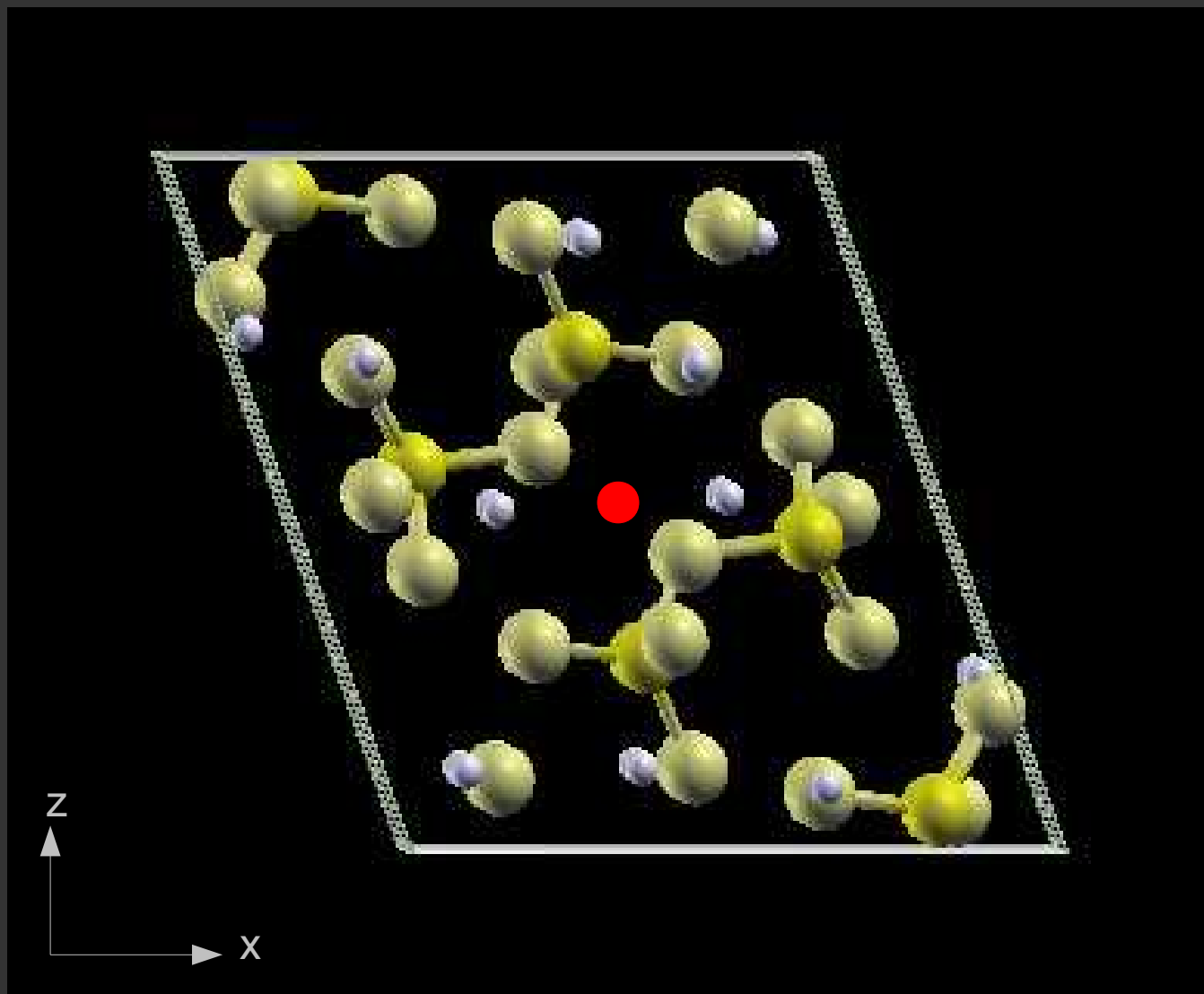
“Thus the superionic $Li_7P_3S_{11}$ was precipitated as a metastable phase in the case of crystallization of glass. The superionic metastable phase is responsible for the high conductivity and low activation energy for conduction of the $70Li_2S \cdot 30P_2S_5$ glass-ceramic.”

-M. Tatsumisago and A. Hayashi, J. Non-Cryst. Solids 354 1411-1417 (2008)



Optimized Crystal Information Unit cell:

- 42 atoms(2 formula units)
- Has inversion symmetry about the red dot
- Contains dimers and tetrahedral sites
- Lattice information
 - Triclinic Cell
 - Sgroup P1
 - X=22.69415500
 - Y=11.64725720
 - Z=23.14655326
(units in bohr)
 - $\alpha=102.462972$
 - $\beta=113.776362$
 - $\gamma=72.1333972$



Computational Methods

- These calculations were carried out using Pwscf. (www.pwscf.org).
- Migration paths were calculated using the Nudged Elastic Band (NEB method) as implemented in Pwscf. (*G. Henkelman, B.P. Uberuaga, H. Jónsson, J. Chem. Phys. 113 (2000) 9901–9904.*)
- Calculations were carried out in a 1x2x1 supercell.
- Calculations were tested with both PAW based and Ultra-soft pseudopotentials.

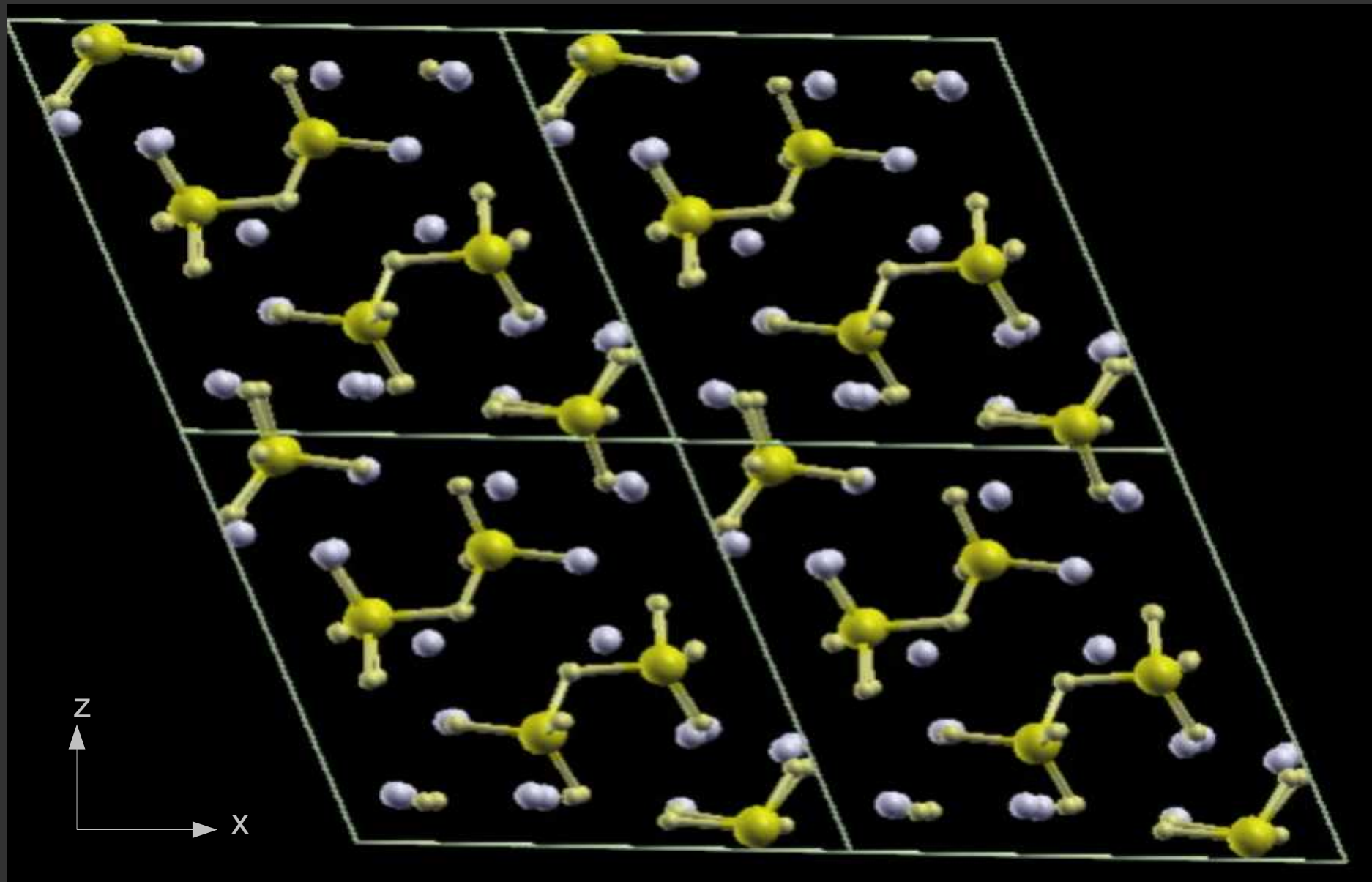
Superionic Conductors

- $\text{Li}_7\text{P}_3\text{S}_{11}$ is a superionic conductor
- Mechanism of ion conduction
 - Vacancy/interstitial pair formation
 - Diffusion through crystallographic voids, both inherent and created.
 - Energy costs
 - Forming pair
 - Moving Li-ions through the lattice.

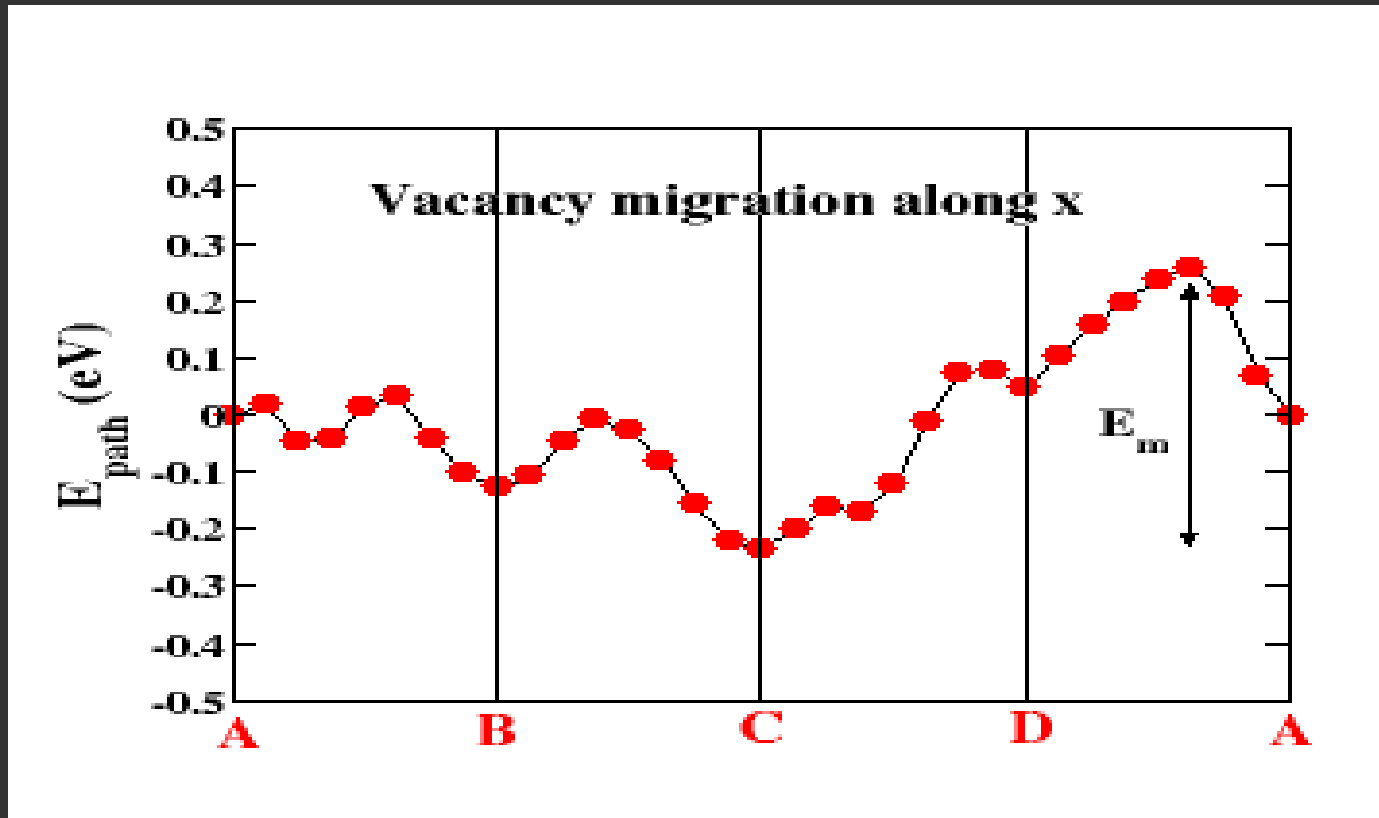
Vacancies

- The crystal contains 7 unique Li positions, and thus 7 unique vacancy sites/energies.
- Using the NEB method, we calculated barriers to migration and migration paths between vacancy sites along all three crystallographic directions.
- The X and Y paths were substantially more energetically favored.

X Path



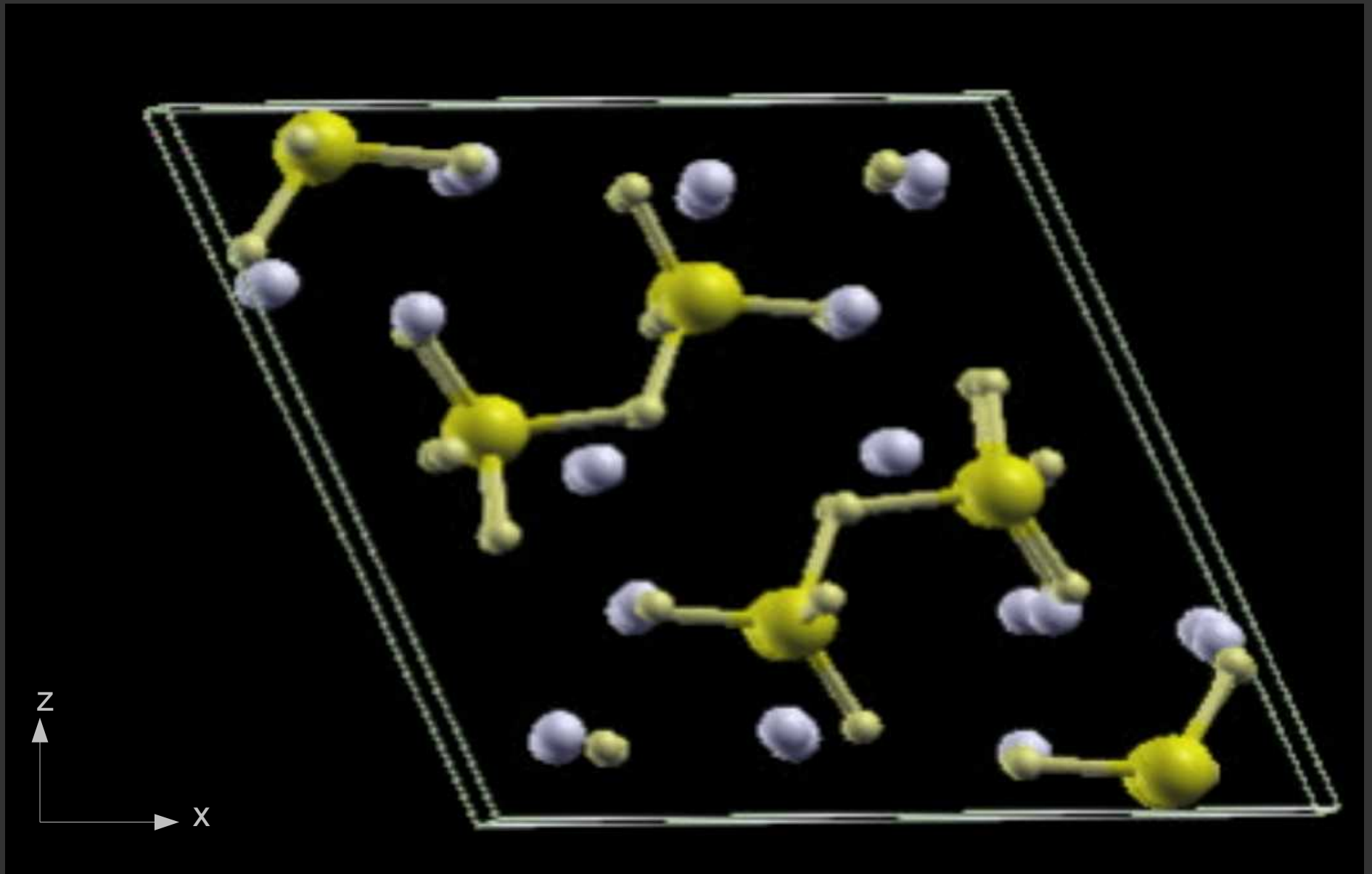
X-direction migration Path



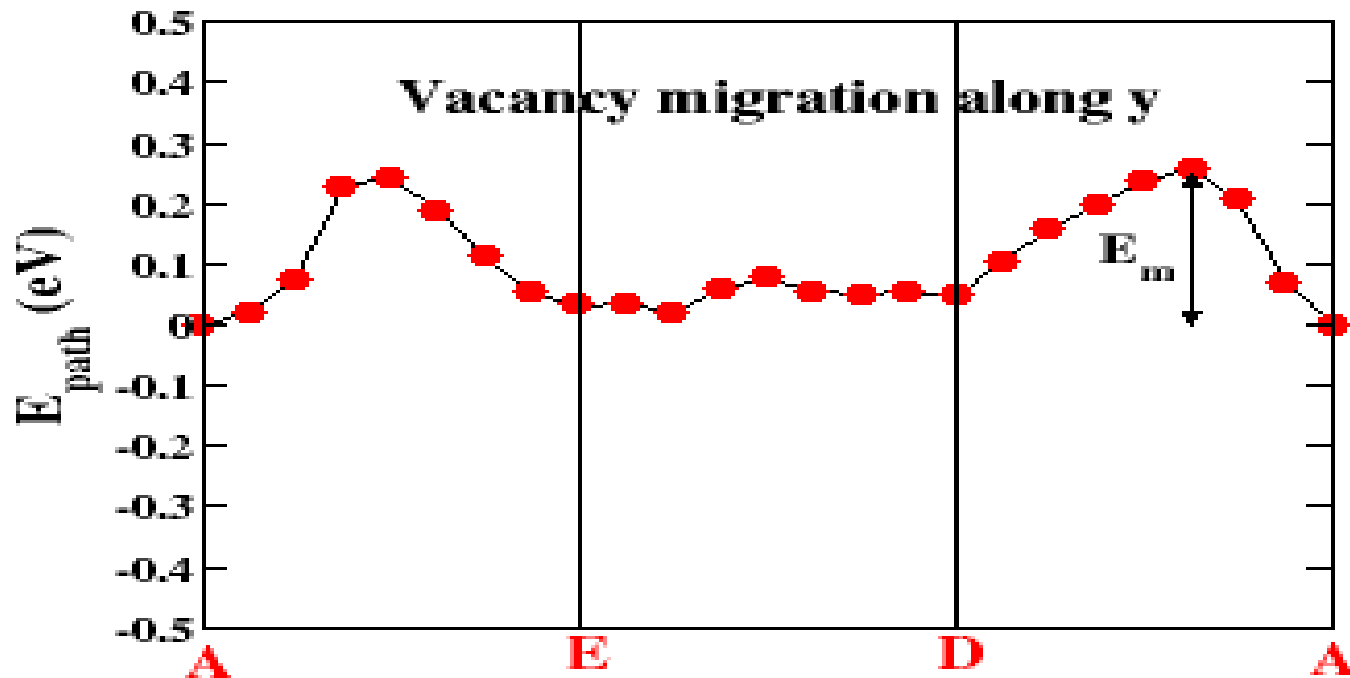
Energies	
A	0
B	-0.13
C	-0.23
D	0.05
E	0.03
F	0.16
G	0.01

- Migration Barrier is $E_m \sim .5$ eV as can be seen above
- Distance travelled in x-direction: 22.7 bohr

Y Path



Y-direction migration Path

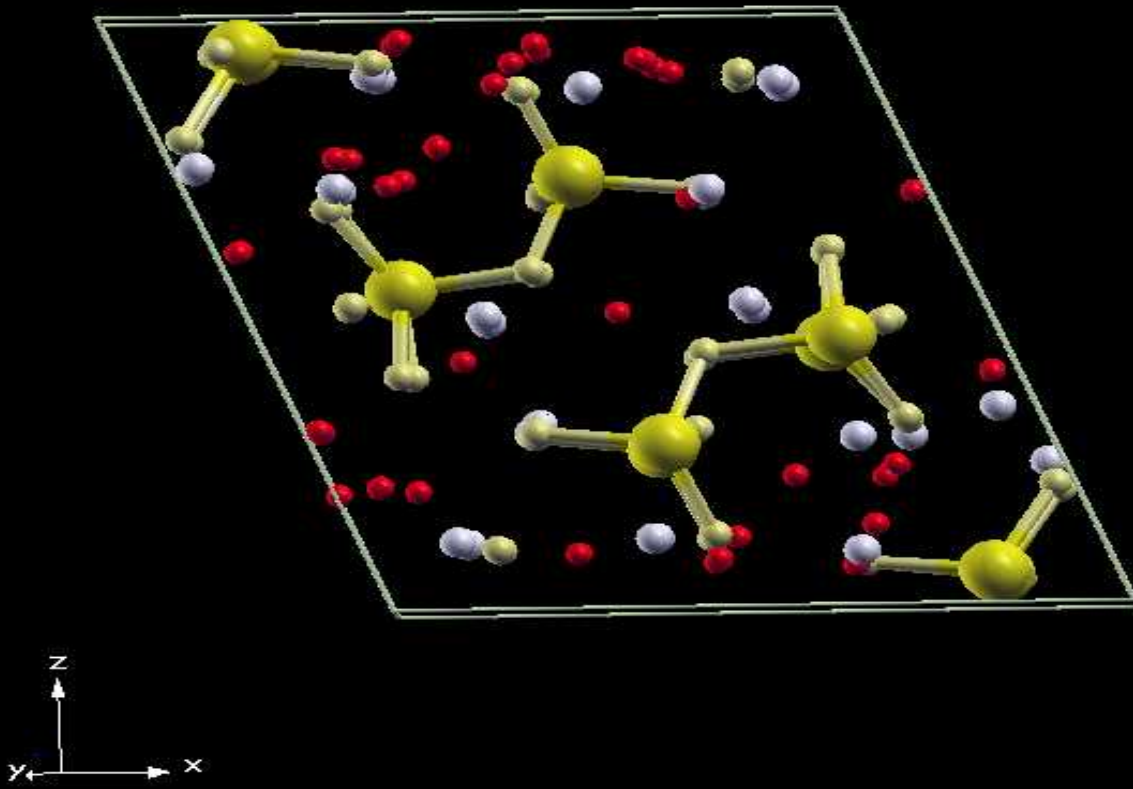


Energies	
A	0
B	-0.13
C	-0.23
D	0.05
E	0.03
F	0.16
G	0.01

- Migration Barrier is $E_m \sim .25$ eV as can be seen above
- Distance travelled in y-direction: 11.65 bohr

Interstitial Sites

A sampling of metastable interstitial sites is shown here in red



Vacancy/Interstitial Formation

- Vacancy/interstitial pairs are created by displacing a Li from a site in the crystal to a metastable interstitial site
- Vacancy interstitial pair formation requires a minimum of $\sim .2$ eV for the sites we have investigated, though our simulations have yet to explore the complete configuration space.

$$\sigma = C \exp \left(-E_m / kT \right)$$

$$E_A = E_m + E_f / 2$$

Conclusions

- We have investigated several Li (thio)phosphates, finding several corresponding stable and metastable structures.
- For the superionic conductor $\text{Li}_7\text{P}_3\text{S}_{11}$ we find vacancy-mechanism driven migration along the y-axis with migration energy $\sim .25$ eV
- More work needed.

References

- Supported by: NSF Grant DMR-0705239
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- P. Giannozzi, S. Baroni, et al., *J. Phys.: Condens. Matter.* 21, 394402 (2009); available from the website: <http://www.pwscf.org/>.
- N. J. Dudney, *Interface* 17 (3), 44 (2008).
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- A. Hayashi, K. Minami, F. Mizuno and M. Tatsumisago, *J. Mater. Sci.* 43 (2008), pp. 1885–1889.
- M. Tatsumisago and A. Hayashi, *J. Non-Cryst. Solids* 354 1411-1417 (2008)